

Chapter 3

Growth Structure and Device
Processing of DQW InGaAs/GaAs
Laser Diode

3. Growth Structure and Device Processing of DQW InGaAs/GaAs Laser Diode

Fabrication of high-power laser diode requires a multilayered structure of semiconductor materials having excellent electrical and optical quality, which is realized by the epitaxial growth. A series of processing steps are required for the fabrication of laser diode bars and devices out of the epitaxially grown laser diode structures. Post-growth processing is a crucial issue for device fabrication and demands very careful optimization, since threshold current and external efficiency depend on internal structures and the device processing of laser diode. This chapter presents a brief report on the laser diode structure and the processing steps used for the fabrication of edge-emitting high-power laser diodes.

3.1 Introduction

Laser diodes have been realized over a wide range of lasing wavelength with a variety of material systems. Since the first demonstration of the GaAs based homojunction laser diode in 1962 [73], the threshold current densities have been brought down by more than three orders of magnitude due to the advancement in the crystal growth technologies. In the early 1970's, improvements in Liquid Phase Epitaxy (LPE) enabled the realization of the double heterostructure (DH) and continuous-wave (CW) operation of the laser diode at room temperature [12]. Further, the development of quantum-well (QW) and quantum-dot (QD) lasers, which are essential for realizing high-power laser diodes today, is possible due to the advancement in molecular-beam epitaxy (MBE) and metalorganic vapor phase epitaxy (MOVPE).

Most high-power laser diodes are based on III-V semiconductor QW structure having an active layer thickness of about a few nanometers, which consists of precisely controlled composition and good homogeneity throughout a wafer. The photon, generated by the band-to-band recombination in the QW active region, is vertically (transversely) confined by means of the waveguide layer comprising of the cladding and confinement layer. The waveguide layers are much thicker, having thickness of 3-5 μm , and in order to assure the economic feasibility of the process they have to be grown at a reasonable

growth rate. All of these have to be achieved while maintaining high crystalline quality with minimal defects, with a tight control of the doping profile, and requiring optimized epitaxial growth conditions for the individual layers of the laser structure. On the other hand, lateral optical confinement is usually achieved by various post-growth processing steps that define the laser device geometry viz. gain-guiding or index-guiding.

3.2 Epitaxial Growth of Semiconductor Materials

A lattice matched single crystal layer with controlled thickness and doping can be achieved by means of an epitaxial growth process. The identical crystalline symmetry between the substrate and the epitaxial layer is crucial for good quality epitaxial growth of the layers. Selection of epi-layer composition to achieve lattice-matching, while providing good carrier and optical confinement, is particularly important for the heterostructures used in laser diodes. The lattice mismatch, i.e. the difference between the lattice constants of the substrate and the epitaxial layer, should not exceed a critical limit. The large value of lattice mismatch usually results in the generation of dislocations and defects, degrading the quality of the grown layer. The lattice-mismatch also determines the onset of three-dimensional (3D) growth-mode, commonly known as the Stranski-Krastanov growth-mode [74].

3.3 Epitaxial Growth Techniques

There are several epitaxial growth techniques viz. liquid phase epitaxy (LPE), molecular beam epitaxy (MBE) and vapor phase epitaxy (VPE), of III-V compound semiconductor materials and their heterostructures. Table 3.1 shows a comparative of various merits and demerits of aforementioned epitaxial growth techniques [75].

3.3.1 Vapor Phase Epitaxy (VPE)

Vapor phase epitaxy (VPE), also known as chemical vapor deposition (CVD), refers to the formation of a thin film on a crystalline substrate from a gaseous medium of different chemical composition. Unlike physical vapor deposition (PVD), where condensation occurs in the absence of a chemical change, the thin film growth takes place in VPE as a

result of thermo-chemical vapour-solid reaction. In VPE, a mixture of gases flow through a reactor and interact on a heated substrate to grow an epitaxial layer. There are three different CVD techniques used for the growth of III-V and other compound semiconductors. These are the halide process, hydride process, and the organometallic or metal-organic vapor phase epitaxy (OMVPE or MOVPE).

Table 3.1: The comparison of epitaxial growth techniques.

Technique	Merit	Demerit
Liquid Phase Epitaxy (LPE)	<ul style="list-style-type: none"> • Simple • High purity 	<ul style="list-style-type: none"> • Commercial production • Limitation to achieve low-dimension
Molecular Beam Epitaxy (MBE)	<ul style="list-style-type: none"> • Uniform layer • Simple process • Abrupt interface • In-situ monitoring 	<ul style="list-style-type: none"> • As/P alloy difficult • 'Oval' defects • Expensive • Low throughput
Hydride Vapor Phase Epitaxy (HVPE)	<ul style="list-style-type: none"> • Well developed • Large scale production 	<ul style="list-style-type: none"> • No aluminium alloys • Complex process • Control difficult • Hazardous sources
Metal Organic Chemical Vapor Deposition (MOCVD)	<ul style="list-style-type: none"> • Most flexible • Large scale production • Uniform layer • Abrupt interface • Simple reactor • High purity 	<ul style="list-style-type: none"> • Expensive sources • Hazardous precursors

❖ Metal-Organic Vapor Phase Epitaxy (MOVPE)

Metal-organic vapour phase epitaxy (MOVPE), also known as organometallic vapour phase epitaxy (OMVPE) or metal-organic chemical vapour deposition (MOCVD) is an improvised form of VPE technique where the growth is carried out from the vapor phase using organometallic and hydride sources [76]. It was first developed extensively by Manasevit in 1968 for the growth of a variety of materials including III-V semiconductors [77]. Since then, it has been advancing as a significant technique for the growth of high-quality low- dimensional structure for fundamental semiconductor physics and for semiconductor photonic devices such as single layers, heterojunctions, and QW structures with excellent control over layer thickness, doping, and formation of hyper-abrupt isotype and anisotype junctions. It was the MOVPE technique, which led to the room-temperature operation of QW lasers for the first time [78] by means of the high-quality epitaxial layer structures.

A typical MOVPE growth system, shown in Fig. 3.1, consists of a gas-mixing system, a reactor, and a pump, and an exhaust handling system. In the gas-mixing system, the metal-organic (MO) precursors are kept in stainless steel container at a precisely controlled temperature and pressure. Flowing hydrogen, carrier gas, through these cylinders of liquid (TMGa, TEGa, TMAI, DMZn) or solid (TMIn) precursors results in a controlled transport of their saturated vapor into the reactor. The reactants can be switched into the reactor or bypassed directly into the exhaust by switching manifolds [79]. The fluxes of hydrogen and the other precursors for epitaxial growth are metered by mass-flow controllers, actually measures the thermal conductivity of the flowing gas. Capacitance or piezoelectric pressure transducers control the pressure. For the reactor design a variety of different concepts are being used, either horizontal linear reactor [80] or vertical reactor [81]. A most common design found extensively for the production of high-power laser diode is horizontal reactor where the gas stream enters on one side, is passed over the lying substrate and leaves the reactor on the other side. The reactor chamber is followed by a pumping system consisting of a rotary pump, a throttle valve for pressure control, and traps for particles or condensable materials. Wet or dry chemical scrubbers are used to remove the toxic materials from the effluent gas coming out of the reactor.

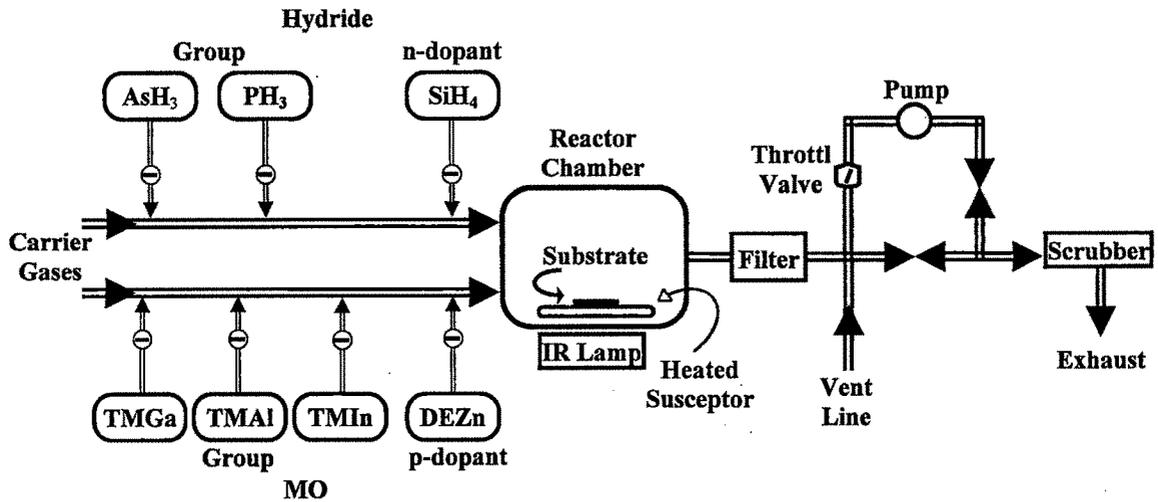


Figure 3.1: A typical MOVPE growth system.

Figure 3.2 illustrates the process in MOVPE reactor during the epitaxial growth. The reaction is carried out in a gas flow with hydrogen, as a carrier gas, and nitrogen, as a pure gas, at atmospheric or reduced pressures. The pure gas decontaminates all the system components from toxic gases and inflammable sources. In this process, chemically active species, or metal-organic precursors, usually the alkyl compounds of group III elements, i.e. Al, Ga or In, interact with sources of group V element, i.e. As and P usually the hydrides arsine (AsH_3) and phosphine (PH_3), on a substrate by means of adsorption, surface diffusion, and surface reaction to produce a corresponding epitaxial layer, either in the vapor phase or on a solid surface of the substrate. The substrate is placed on a heated graphite susceptor. The hot susceptor has a catalytic effect on the decomposition of gaseous products and the growth takes place on this hot surface. The byproducts are removed through diffusion.

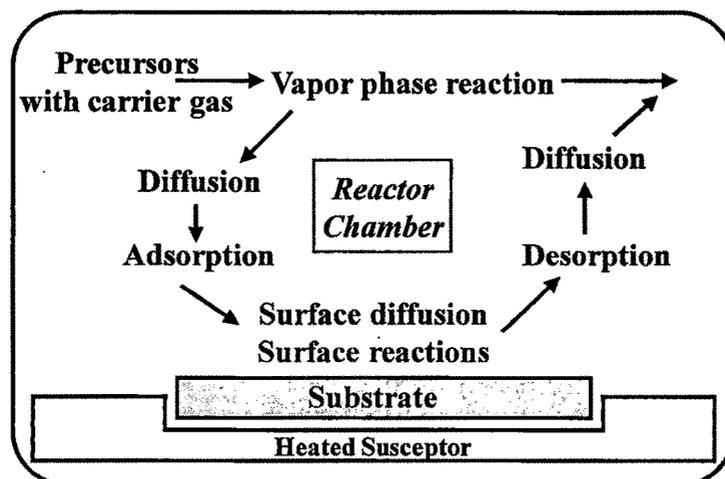
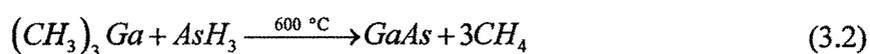


Figure 3.2: Processes in MOVPE reactor during the epitaxial growth.

In MOVPE growth process, the basic chemical reaction is irreversible pyrolysis that takes place in the vapor phase of a volatile metal-organic compound and a gaseous hydride, and forming some intermediate compounds, given by, Eq. 3.1 [76]



where, R_n indicates a lower order organic radical, such as a methyl- or ethyl-radical; A and D are, the group III and V constituent species for the deposited solid, respectively. An important example of this case is given by, Eq. 3.2,



Finally, the constituent elements are released and get incorporated into the lattice on the semiconductor surface.

The growth rate is one of the main parameters to be controlled during epitaxial process, and is mainly decided by the growth temperature. The kinetics and thermodynamics of the growth process play an important role, which defines the growth rate and determines the driving force for the overall growth process, respectively. If the reaction rate limits the growth rate, it is called kinetically limited case wherein the growth rate increases with increase in temperature. Moreover, the processes like diffusion, called mass transport, too, limit the growth rate, which is inversely proportional to pressure and is independent of temperature. Growth rate is usually controlled by reaction kinetics in low-temperature range and by mass transport in high-temperature range. The growth rate is given as $r_g = (\text{const.}) p_{TMGa}(V/P)^{1/2}$ for the growth process of GaAs, as in Eq. 3.2, where, p_{TMGa} = partial pressure of TMGa, V = gas velocity, P = reactor pressure.

A number of important and beneficial features are offered by the MOVPE growth technique. Compounds, such as GaAs, can be grown with different As to Ga ratio (V/III ratio) simply by varying the relative vapor pressure of arsine hydride (AsH_3) and trimethyl gallium (TMGa) as there is no need of establishing equilibrium with source materials. This allows the study of native defects and trap states, which are supposed to be related to the stoichiometry. It is possible to grow almost all the III-V semiconductor compounds, their ternary and quaternary alloys, using MOVPE. Growth of high purity, ultra-thin layers, and abrupt junctions with precise composition and doping control makes this technique more prevalent in the field of optoelectronic device fabrication.

3.3.2 Materials for High-Power Laser Diodes

The direct energy bandgap and lattice constant of the material are the most crucial parameters in the material selection for laser diodes. Hence, the elementary semiconductors e.g. silicon or germanium are not used for lasers because of their indirect energy bandgap. The III-V compound semiconductor materials are widely used for the fabrication of high-power laser diodes. The III-V binary compounds are formed by combining elements from group III and group V of the periodic table while, ternary or quaternary compounds are formed through alloyed binary compounds. The energy bandgap and the lattice constant of ternary and quaternary compounds depend on the mole fraction of the constituent binary compounds. For example, the lattice constant of any ternary compound ($A_xB_{1-x}C$) is calculated with the help of Vegard's law [82] using Eq. 3.3.

$$a_{A_xB_{1-x}C} = xa_{AC} + (1-x)a_{BC} \quad (3.3)$$

where, AC and BC are binary compounds such as InP, GaP. Similarly, the energy bandgap of any ternary compound ($A_xB_{1-x}C$) can be calculated as,

$$Eg_{A_xB_{1-x}C} = xEg_{AC} + (1-x)Eg_{BC} + x(1-x)C_I \quad (3.4)$$

where, C_I is the Bowing parameter of ternary material [83].

As the alloy composition changes the band structure and hence electronic and optical properties of the alloy changes. Thus by choosing appropriate compounds or alloys, it is possible to select and tune the emission wavelengths for laser diodes. The choice of material system for the fabrication of laser diode is determined by the emission wavelength (λ) of interest and hence by the energy bandgap (E_g) of the active region in the laser structure as,

$$E_g \text{ (eV)} = \frac{hc}{\lambda} = \frac{1239}{\lambda \text{ (nm)}} \quad (3.5)$$

where, h is Plank's constant and c is velocity of light. Therefore, the lasers having wavelength in the range of 600-700 nm can be assessed using ternary or quaternary compound viz. AlGaAs, InGaP, AlGaInP. Also, lattice matched AlGaAs grown on GaAs can realize laser emission wavelength in the range 700-870 nm, whereas using InGaAs/GaAs strained quantum well system the wavelength range can be extended

further up to 1100 nm by varying the thickness and indium content. The energy bandgap and lattice constant for various semiconductors at 0 K [84] is shown in Fig. 3.3.

Laser diode often faces rapid degradation in its performance due to formation of misfit dislocations in the crystal. The density of point defects, which act as centers for non-radiative recombination and reduces the efficiency, should be as low as possible. To avoid this problem, thick waveguide and cladding layers have to be grown lattice matched to the substrate. Moreover, the crystalline quality of the layers can be adversely affected by ordering as in the case of GaInP or by phase separation as observed for certain compositions of quaternary GaInAsP [81]. These require optimal growth conditions for each material.

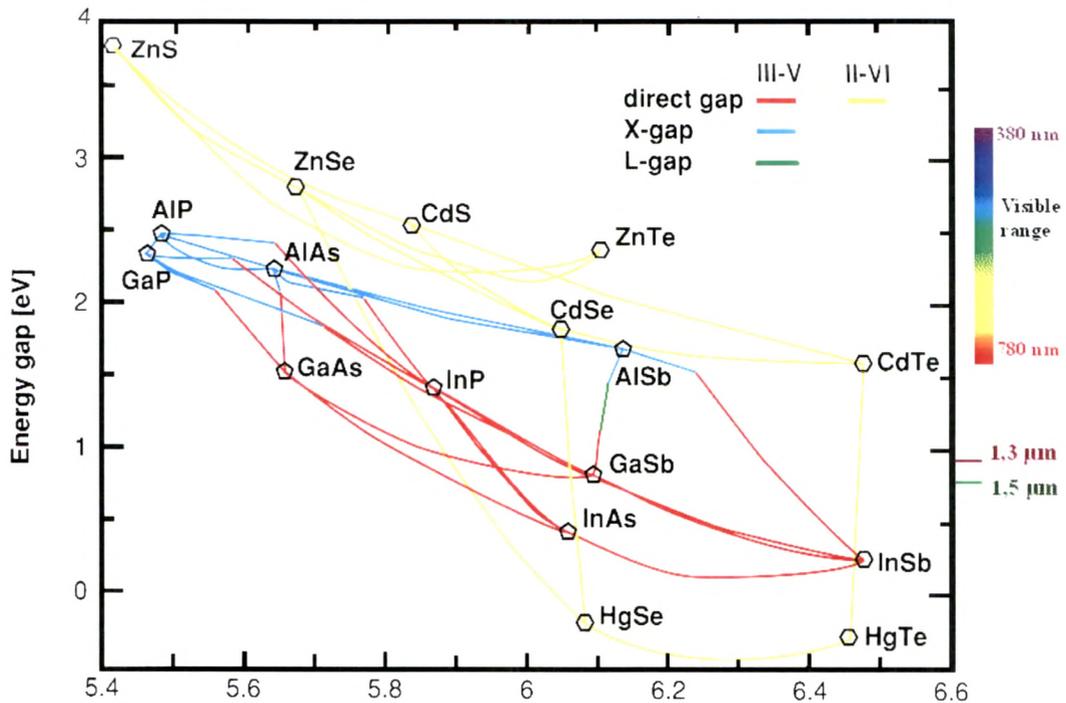


Figure 3.3: Bandgap versus lattice constant for conventional III-V semiconductors at 0 K.

3.3.3 Growth of a Complete Laser Structure

After optimizing individual layer parameters, complete laser structure has been grown by MOVPE (AIX-200) at RRCAT Indore [37]. The laser structure was grown for the fabrication of 980 nm high-power laser diode. Figure 3.4 (a) shows schematic of

complete laser structure with thickness of each layer. The energy band diagram corresponding to the laser diode structure is shown in Fig. 3.4 (b).

Further processing of this MOVPE grown laser was carried out for the structural growth of laser diode. The device processing is crucial in the high-power laser diodes. Since the high-power operation of the laser diode involves high-current density and high light intensity, the processing procedures must be optimized for these devices to achieve the following viz. (i) to improve the optical and electrical confinement, (ii) to minimize the series resistance of the device and, (iii) to reduce the amount of stresses and damages experienced by the laser diode structure during the processing steps.

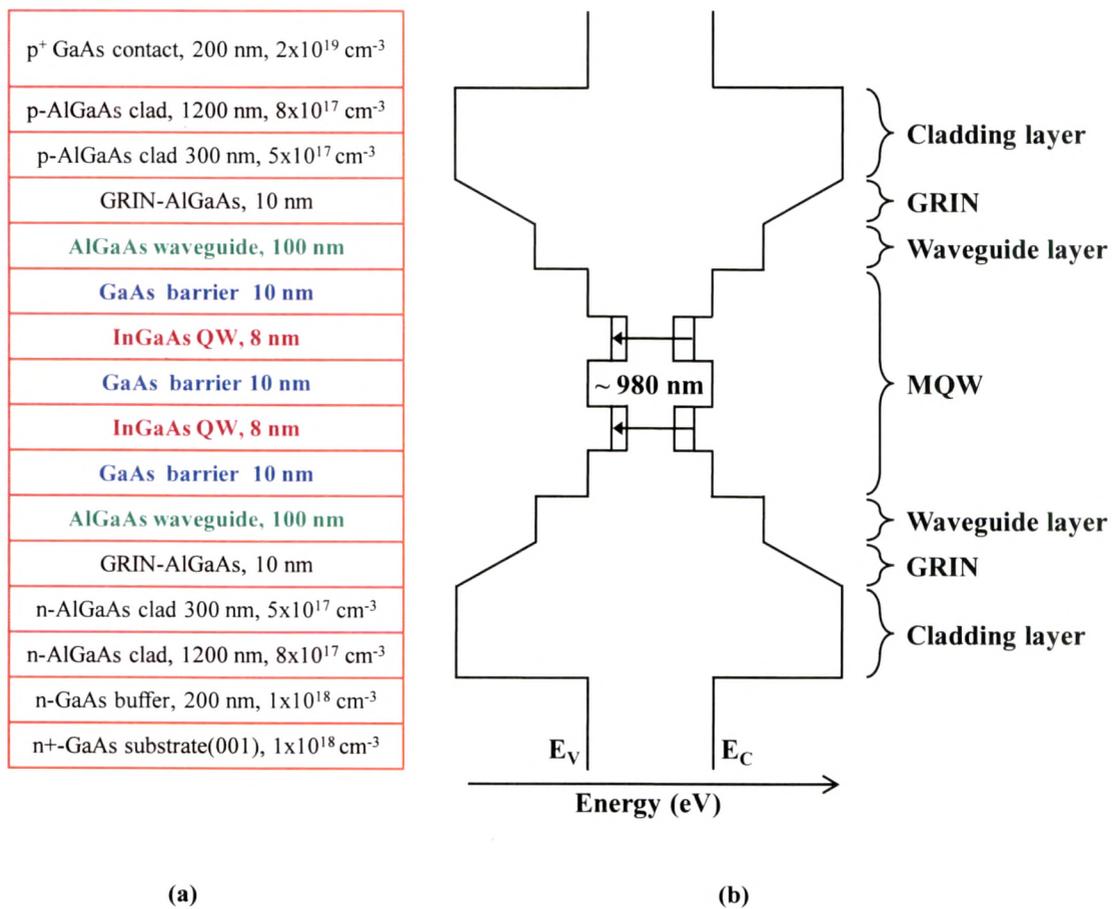


Figure 3.4: (a) Schematic structure for 980 nm high-power laser diode. (b) Energy band diagram for the grown laser diode structure.

3.4 Various Laser Diode Geometries

The edge emitting laser diodes are fabricated mainly in two device geometries. These are broad-area geometry and stripe geometry.

3.4.1 Broad-Area Laser Diodes

The broad-area laser diodes are the most elementary form of edge-emitting laser diodes. In these lasers, the stripe-width is much larger than the thickness of the laser structure, which allows uniform distribution of the injected current and the optical field in lateral direction. Nevertheless, due to large lateral dimensions many transverse modes can sustain this type of laser structure. Moreover, the filamentation, i.e. lateral beam distortions due to increase in refractive index, in broad-area devices is very well-known, and limits the high-power operation of the device.

3.4.2 Stripe Geometry Laser Diodes

As the name suggests, laser diodes with a few μm wide narrow stripe for lateral current confinement are known as stripe geometry injection lasers. This current confinement into the device can effectively overcome the issues like filamentation. The stripe geometry laser diodes can again be categorized into gain-guided, index-guided and buried heterostructures, according to different types of lateral confinement mechanisms viz. current, optical or photon, and carrier confinement, respectively.

Laser diodes, which only have current confinement, are called gain-guided lasers. In case of gain-guided lasers the lateral wave-guide is defined simply by the contact geometry. The current confinement is provided by an aperture, which is mostly realized by a dielectric isolator as shown in Fig. 3.5 (a). Though the fabrication of this laser is easy, there are some disadvantages as well. The threshold current in this devices is high compared to the index-guided one because of optical loss in waveguide.

The waveguiding in index-guided laser diode is achieved by means of lateral effective refractive index step difference formed by a very small mesa-structure in the

thickness of the upper cladding layer. A single lateral mode operation can be attained by optimizing the index step and the width of the waveguide. A typical example of an index-guided laser is the mesa- structure or ridge-waveguide laser shown in the Fig. 3.5 (b). Since, the current is injected at the top of the ridge, current confinement is also realized. A precise and reproducible control of ridge dimensions is necessary during device fabrication as the beam quality of the ridge-waveguide laser strongly depends on ridge dimension. The main advantage of the mesa- structure or ridge waveguide is well-defined current and optical confinement, which results in a lower threshold current, especially for stripe widths smaller than $100\ \mu\text{m}$.

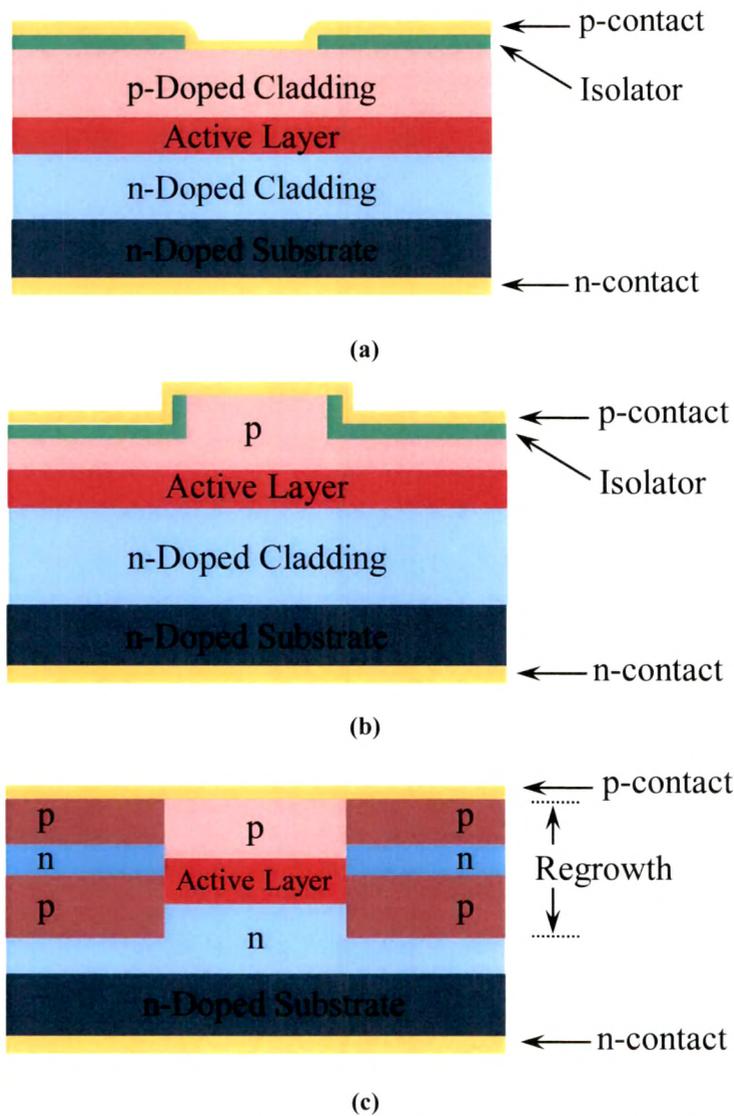


Figure 3.5: Stripe laser structures with different lateral confinement. (a) A gain-guided stripe laser, (b) The ridge-waveguide laser and (c) The buried-heterostructure laser.

The third type of stripe-geometry laser diode is a buried heterostructure, having all three types of lateral confinement, shown in Fig. 3.5 (c). The buried heterostructure is composed by an epitaxial re-growth technique, provides index guiding and carrier confinement. The p-n-p structure in the vertical direction works as a current-blocking layer providing carrier confinement. Buried heterostructure lasers are mostly used in communication systems where low- power consumption is required.

The stripe geometry laser provides no optical confinement and has large optical losses hence higher threshold current. On the other hand, the buried heterostructure, providing low threshold current indeed, is quite difficult to fabricate and need to optimize precisely for its reproducibility. Hence, the ridge-waveguide laser structure having both current and optical confinement with moderate threshold current and low optical loss is easy to optimize. We have processed the epitaxially grown wafer to grow mesa structure ridge waveguide laser. The optimization of all processing steps is discussed in detail in following section.

3.5 Fabrication of Laser Diode Bar with Mesa-Stripe Geometry

After the MOVPE growth of laser diode structure, the epitaxially grown wafer was processed to make laser diode bars using mesa-stripe geometry. For the device processing, the grown wafer is first cleaned by means of organic cleaning. The next step is to form the top ohmic contacts. The metal patterning in laser diode processing is usually attained by the photolithography and lift-off process. The photolithography involves applying an organic compound, called photoresist, on the wafer followed by baking, ultraviolet (UV) exposure through a pattern mask, and development of pattern to determine the desired exposed patterns. The device-isolation to form mesa-structure is usually carried out by wet chemical etching and coating of dielectric oxide e.g. SiO₂. The wafer is etched down to the appropriate layer of multilayered laser diode structure in the unmasked areas.

Further, the laser diode wafer has to be thinned down to desired thickness by lapping and polishing of substrate. Finally, the laser diode bars with desired cavity length

are scribed and cleaved. The processing is generally carried out in the class-10,000 clean-room, i.e., the room contains less than 10,000 particles of size more than 0.5 μm in a cubic foot of air. All these steps for fabrication of laser diode bars with mesa stripe geometry and their optimization for our structure is discussed in the next section in their processing sequence.

3.6 Optimization of Various Steps in Laser Diode Processing

The post-growth device processing of laser diodes were carried out at Semiconductor Laser Section, SSLD, RRCAT, Indore. The wafers were first cleaved into small pieces of about 1 inch area. Here, it is important to identify the crystalline direction lying within the surface plane since the etching profile is good in the primary flat direction and it is easy to cleave the sample in the direction perpendicular to the primary flat in order to obtain the cavity mirrors.

3.6.1 Organic Cleaning

The organic cleaning is preliminary practice to clean the semiconductor wafer from various contaminants e.g. dust, greases and other oily impurities. The sample was first cleaned in warm trichloroethylene (TCE) which removes oily impurities from the sample. Furthermore, the sample was cleaned in acetone and methanol, respectively, to remove the effect of previously used chemical, i.e. acetone to remove TCE and methanol for acetone. Finally, the sample was cleaned by de-ionized (DI) water. These organic solvents are effective in removing oils, greases, waxes and organic materials such as photoresists from the sample surface. The sample was then dried under nitrogen flow, and made ready for photolithography, which defines the transverse dimension of laser diode.

3.6.2 Photolithography

Photolithography is a process to form desired pattern on semiconductor surface by means of pattern-masking on an intermediate photosensitive polymer film called photoresist. The semiconductor surface, coated with photoresist, is then exposed through mask and developed to realize desired pattern by removing excess photoresist. The remaining

pattern can then be replicated in other materials in subsequent processing, using techniques such as etching, metallization, etc. The resolution is limited by the diffraction effect, which increases with the square root of the wavelength and with the gap between the mask and the wafer [85]. Various steps involved in photolithography are given below.

❖ Spin Coating

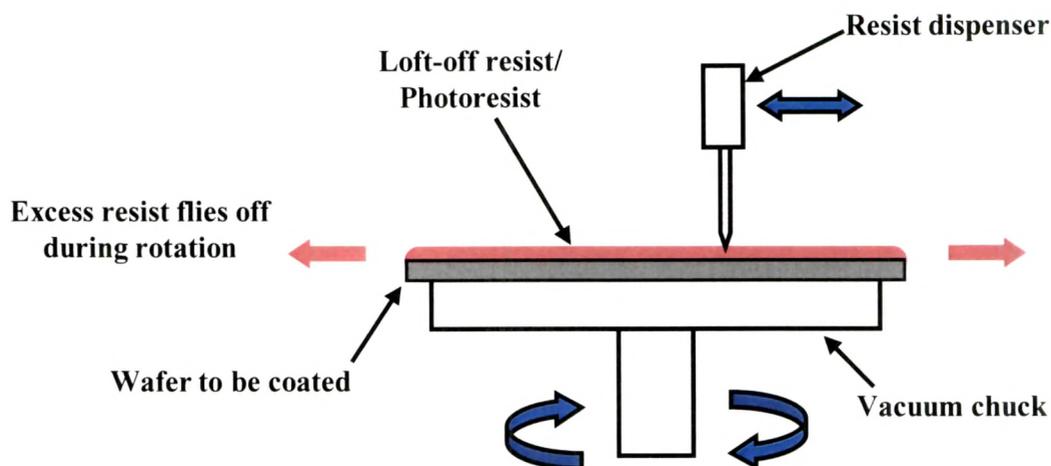


Figure 3.6: Schematic diagram of Spin coating process

➤ Lift-off Resist Coating

Prior to photoresist coating, the lift-off resist (LOR) has been spin-coated for the subsequent lift-off process to remove excess metal coat on the photoresist. It is used for making undercut in the photoresist layer. Since, LOR do not contain any photosensitive material it does not decompose on exposure to UV light. The LD samples were spin-coated with LOR at 4000 rpm for 30 sec following to the backing for 5 minute at 150 °C.

➤ Photoresist Coating

Generally, photoresist materials are classified into two types according to their solubility in a developer after UV exposure, i.e. positive photoresist (PPR) and negative photoresist (NPR).

- **Positive photoresist material**

The positive photoresist (PPR) becomes more soluble in a developer after exposure to the ultraviolet (UV) light. It consists of two parts, resin and photoactive compound. It decomposes on exposure to the UV light, through patterned mask, and makes the resin

more soluble in developer. Hence, the exposed resist is then washed away, leaving windows of underlying material.

- **Negative photoresist material**

As the name suggest, the negative photoresist (NPR) behaves exactly opposite to the PPR. It becomes less soluble in developer after exposure to the UV light. It consists of chemical inert rubber and photoactive agent. Exposure to the UV light causes the negative resist to become polymerized, and more difficult to dissolve in developer. Hence, developer solution can only remove the unexposed photoresist. Masks used for NPR, therefore, contain the inverse (or photographic "negative") of the pattern to be transferred.

Both PPR and NPR have their merits and demerits, listed in Table 3.2. We have used PPR for lithography, as it provides good step coverage and dissolve in aqueous developer. It is deposited on the semiconductor surface by means of spin coating.

Table 3.2: Differences between positive photoresist and negative photoresist.

NPR	PPR
Swells during develop	No swelling during develop
Marginal step coverage	Good step coverage
Organic solvent developer	Aqueous developer
Sensitive to O ₂	Operate well in air ^[86]

We used OiR-960 positive photoresist (S1813) for photolithography of laser diode samples. The samples were coated with PPR, having thickness about 1 μm, at 3500 rpm and soft baked at 95 °C on a hot plate for one minute.

❖ **Alignment & Exposure**

A photomask consists of desired pattern, i.e. alternate transparent and opaque stripes, to realize laser diode mesa structure was aligned to the wafer. The photomask, having stripe width 100 μm, used for patterning is shown in Fig. 3.7, having opaque Fe₂O₃/chromium stripe pattern on a soda lime glass substrate.

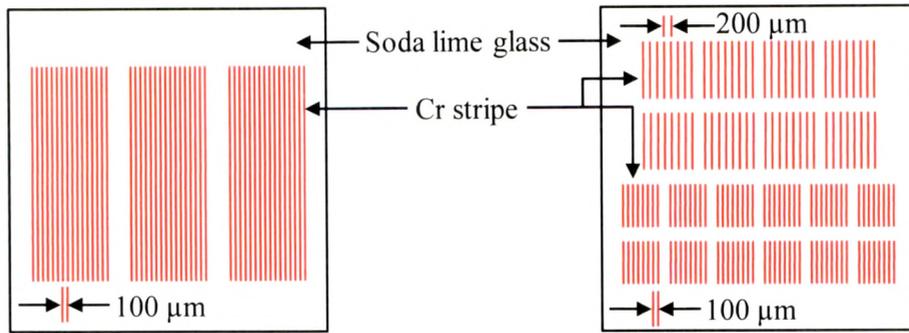


Figure 3.7: Masks used in lithography

In optical lithography, the formation of images with visible or UV radiation occurs on the photoresist using contact or proximity printing. We have used contact printing, in which the mask (M1-MESA (N)) was brought into contact with a resist coated wafer. This prevents the diffraction losses and we get high resolution. However, there is a drawback of this method in that it may produce defects in both, the wafer and the mask. The sample was exposed using a Quintel 4TL mask aligner for 10 s with 10.6 mw, 365 nm UV radiation.

❖ Development

The exposed photoresist is now dissolved into a suitable developer which is different for different photoresists. When the wafer is dipped into a developer the exposed photoresist dissolves in developer (MFCD-26) while lower LOR remains unaffected. After exposure and development, a hard bake at 120 °C for 1 minute is applied to the sample, which makes photoresist hard and prevents the further dissolution of the photoresist in developer. Once again the sample is dipped in MFCD-26 for 1 minute to realize undercut by removal LOR. Finally, the wafer cleaned in DI water and dried with N₂ gas

3.6.3 Top Metal Contact Deposition

The purpose of top and bottom metal contact is to provide an ohmic contact to the laser diode which allows the flow of electric current through the device. Hence, the series resistance of the device due to contact layer should be as small as possible. Also, the contact layer should be a stable with respect to ambient temperature and over the long

period time. Therefore, low contact resistance, ease of fabrication, better adhesion, and thermal stability are the primary quality requirements of the metal to realize a good quality contact layer.

The metallization is also the basis for the laser diode mounting on a heat sink which allows soldering, also known as die-bonding, and wire-bonding. Generally, for high-power diode lasers based on GaAs substrates, the semiconductor contact layer consists typically of heavily doped *p*- and *n*-type GaAs. Almost any metal placed in intimate contact with these heavily doped GaAs surface will result in an ohmic contact without any alloy formation. We have deposited multilayer of Titanium (Ti) (~10 nm)/ Platinum (Pt) (~5 nm) / Gold (Au) (~200 nm) in sequence to deposit p-type ohmic contacts. Here, Ti and Pt are used for the better adhesion of top Au layer, which provides a very high conductivity and a soft surface quality for device bonding. The front contacts to the p+ GaAs capping layer of the laser structures were deposited using a metal coating unit with thermal evaporation system at high vacuum (3×10^{-6} mbar pressure)

3.6.4 Lift-off Process

Lift-off process is a simple method for patterning films which are deposited on the sample. In this step, we remove the metallic layer from the top of mesas, by means of warm (65 °C) PG remover bath. The PG remover reacts with LOR and causes swelling and losing adhesion to the wafer and hence excess metal deposited on the photoresist is removed while metal directly deposited on wafer remains unchanged. Figure 3.8 shows the final structure of the top p-contact surface.

Any deposited film can be lifted-off provided that the substrate does not reach the temperature high enough to burn the photoresist during the deposition and the film is thin enough to allow solvent to seep underneath. Various mechanisms from resist coat to lift off are shown in Fig. 3.9 [87].

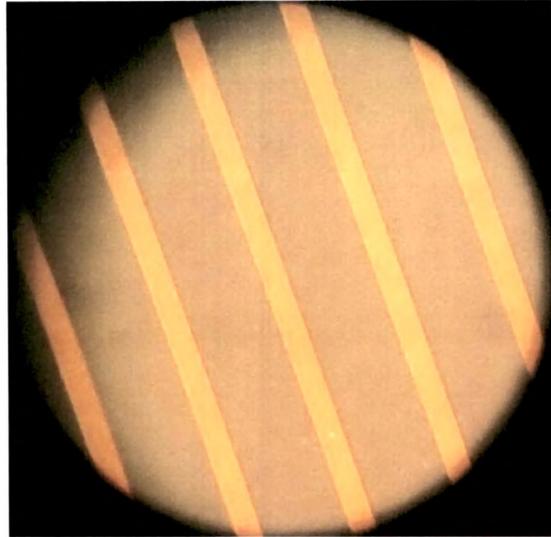


Figure 3.8: Photograph of metal contacts on laser structure after lifting-off.



Figure 3.9: Lift-off technique: (a) overlying layer of photoresist (OiR 960), soft baked at 95 °C for 1 minute and (b) exposed to UV light ($\lambda = 365$ nm) (c) developed resist with formation of undercut (d) metal deposition on sample (e) lift-off and clean.

3.6.5 Rapid Thermal Annealing of Top Contact

Rapid thermal annealing (RTA) is very crucial for obtaining good metal-semiconductor ohmic contacts [88]. It helps the diffusion of the metal into the GaAs crystal, and reduces the barrier-height at the junction by alloying the semiconductor and the metal at the junction. This results in the reduction of series and thermal resistance of the devices. The annealing should be very rapid to avoid unwanted processes like redistribution of impurities and increment in escape probability of volatile materials. The RTA also provides good adhesion for metal to semiconductor contacts.

Heating by an array of incandescent lamps is a popular method for RTA. In this method, many incoherent lamp sources like tungsten-halogen lamps are mounted in a cylindrical box containing a quartz tube in which samples are placed on the Silicon wafer. Silicon wafer has high absorption for infrared radiation, which is emitted by tungsten halogen lamps and hence causes the heating of the sample up to desired temperature. We

have carried out RTA at 375 °C for 45 s. Rapid heating of a sample with temperature and desired dwelling time is a difficult job and this is achieved by a computer controlled PID temperature controller. A thermocouple is used to measure the temperature of wafer. For abrupt cooling, proper coolant like N₂ gas is used. The temperature profile graph of RTA process is shown in Fig. 3.10.

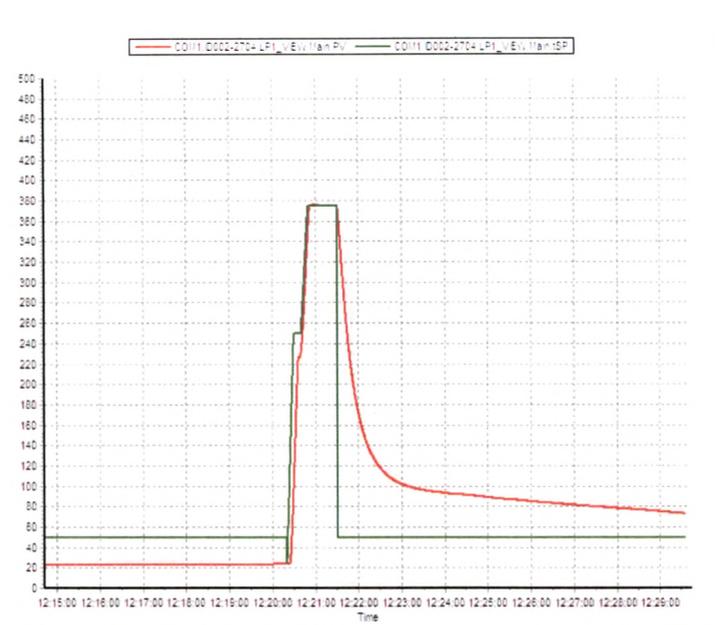


Figure 3.10: Temperature profile with respect to time during RTA process.

3.6.6 Etching

The etching is required to isolate individual devices in a laser diode bar and to form mesa structure. The wafer is etched till the upper cladding layer by using suitable etchant which reduces the current spreading in the devices. The etching of the mesa can be carried out either by wet etching or dry etching methods. In wet etching, areas of the wafer outside the desired stripe are removed by dissolving them in a wet chemical solution. The dry etching involves reaction of materials in such areas, to be removed, with gases in plasma to form volatile products. In this work, because of its simplicity [89,90], and depth control, wet etching was preferred giving reliable results for laser diode devices.

In the case of wet chemical etching, only solutions which exhibit anisotropic etching and strictly proceed downwards into the material are preferred. If the etching is

isotropic, a large undercut occurs resulting in breaks and degradation of metallization films and the structure [85]. Though the wet etching forms an undercut and hard to control it is mostly used due to its low cost, easy to implement, high etching rate, good selectivity for most materials and produce better electronic properties.

The etching solution for GaAs and AlGaAs usually contain hydrogen peroxide (H_2O_2), to oxidize the semiconductor surface, and phosphoric acid (H_3PO_4), to dissolve the oxidized components of mixture. In the present study, H_3PO_4 etching solution has been used. The $\text{H}_3\text{PO}_4 / \text{H}_2\text{O}_2 / \text{CH}_3\text{OH}$ system is used to etch GaAs and AlGaAs at controlled rates. This is due to high viscosity of H_3PO_4 , so that diffusion of the hydrogen peroxide to the semiconductor surface becomes the rate - limiting step. In this study, a mixture of $\text{CH}_3\text{OH} / \text{H}_3\text{PO}_4 / \text{H}_2\text{O}_2$ in 6:3:1 ratio is made as an etching solution for GaAs and AlGaAs. The etching rate was determined with the help of surface profilometer by etching on the dummy samples and desired etching depth was achieved by controlling time of etching, shown in the Fig. 3.11. After etching, the wafer was dried by N_2 gas.

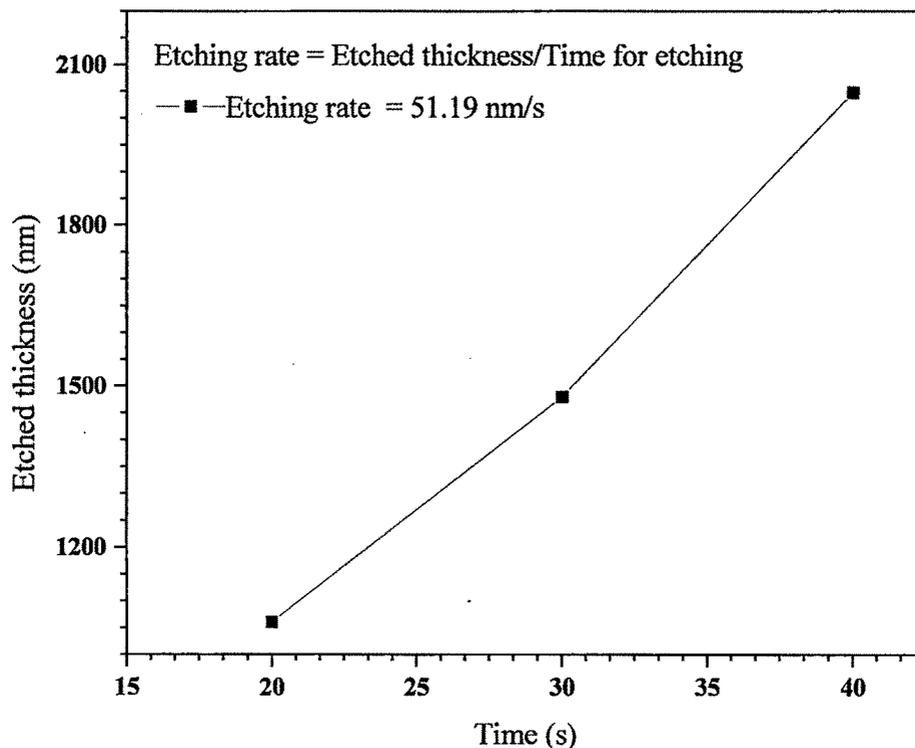


Figure 3.11 - Etched thickness vs. time

3.6.7 SiO₂ Deposition and Window Opening

For better isolation and surface state passivation of laser diode, dielectric layers are used as a current barrier between isolated laser diodes after GaAs/AlGaAs etching. The oxide deposition improves the performance of the laser diode because it isolates the devices completely and prevents tunneling of carriers from one diode to other. So the carrier loss is reduced and directly injected to the active region. Thus the injection efficiency of the laser diode is also improved. We have deposited SiO₂ dielectric layer of ~200 nm on GaAs/AlGaAs etched mesa laser diode structure. After SiO₂ deposition it is necessary to remove the SiO₂ from the top of metal contact region. To do so we have again used photolithography and selective etching of SiO₂ with following sequential process execution.

- 200 nm thick SiO₂ deposition on etched laser diode wafer.
- Spin coating of PPR for 30 s at 3500 rpm.
- Soft baking at 95 °C for 1 min.
- Mask alignment in such a way that only exposed PPR coated on metal stripe.
- Exposure to UV light for 10 s.
- Etching of selective SiO₂ Buffer Hydro-Fluoric acid (BHF).
- Develop for 1 min with developer, to remove PPR from SiO₂.
- Then clean with acetone, methanol, and DI water.
- Dry it with N₂ gas.

To make the Buffer Hydro-Fluoric acid (BHF), 40 g of ammonium fluoride (NH₄F) is dissolved in 60 ml of DI water. The solution is then mixed with HF acid in a ratio of 6:1. The overall chemical reaction for SiO₂ etching is: $\text{SiO}_2 + 6\text{HF} \rightarrow \text{H}_2\text{SiF}_6 + 2\text{H}_2\text{O}$, where H₂SiF₆ is water-soluble complex. The pattern sample was dipped into the BHF solution for 15-20 s for etching. After SiO₂ etching and complete lithography process we get SiO₂ window open from the metal contact region as shown in Fig. 3.12.

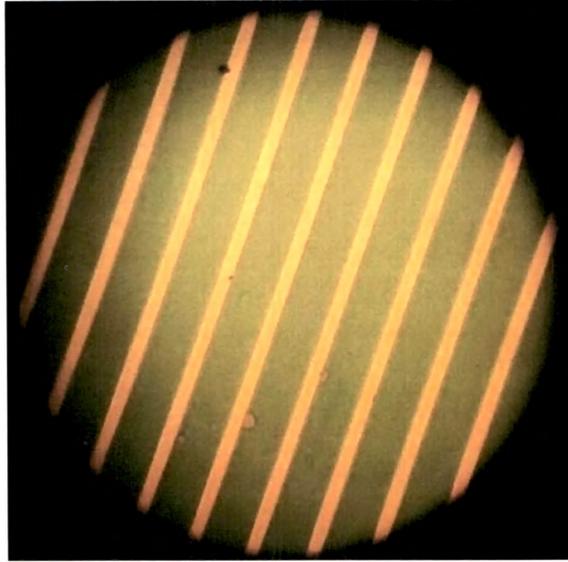


Figure 3.12: Sample photograph after window opening by removal of SiO₂ with BHF.

3.6.8 Lapping and Polishing

In order to reduce the thermal resistance of the device and to facilitate the scribing procedure, it is necessary to thin down the wafer by reducing the substrate thickness. This is accomplished by reducing the thickness of n⁺ GaAs wafer to around 150 μm by mechanical lapping and polishing the wafer manually. Lapping was done using alumina powder or diamond pest, having particle/grain size 5 μm and 1 μm, to reduce thickness followed by polishing with alumina slurry, having particle size 0.5 μm, to mirror polished the surface. Polishing is necessary to achieve uniform surface with minimum roughness which reduces the device resistance and helps to get void-free die-bonding. After lapping and polishing, the wafer was cleaned organically and the bottom surface of the wafer was chemically polished by means of etching using CH₃OH / H₃PO₄ / H₂O₂ etching solution. Finally, the wafer was cleaned with DI water and dried under nitrogen flow.

3.6.9 Bottom Metal Contact Deposition

After the samples are thinned by lapping and polishing, it is necessary to cover this surface entirely, for the n-contact of the laser devices. The most common approach of fabricating ohmic contacts on n-GaAs is to apply an appropriate metallization to the wafer, and then alloy the metal into the GaAs. During the alloying and cooling period, a

component of the metal enters into the GaAs and highly dopes the surface layer. This doping decreases the Schottky barrier and the tunneling dominates the conduction mechanism. The doping agent is generally chosen to be germanium (Ge). Gold-germanium (Au-Ge) is usually applied with an overlay of another thin metal layer such as nickel (Ni). Addition of Ni to Au-Ge leads to lower contact resistance as well as serving to maintain a smooth surface morphology after alloying the contact metallization [91]. The Au-Ge alloy has a poor sheet resistance, is very difficult to wire-bond and is not solderable to a heat sink. This requires an additional gold layer to be deposited on the Au-Ge alloy. The sequence for bottom ohmic contact deposition on n-type substrate is Au-Ge (~5 nm) /Ni (~5 nm) /Au (~100-200 nm). To achieve intimate contact formation between GaAs substrate and the deposited metals, we have carried out RTA at 375 °C for 45 s.

3.6.10 Scribing

Finally, the wafer is cleaved along the cleavage plane, i.e. (110) for GaAs, perpendicular to laser diode stripes using Micro Suss HR 100 manual scriber to get laser diode bars with different cavity lengths ranging from 0.5 mm to 2 mm. Each bar contains several laser diodes with stripe width of 100 μm.

The complete process layout for the processing of laser diode is given in Fig. 3.13.

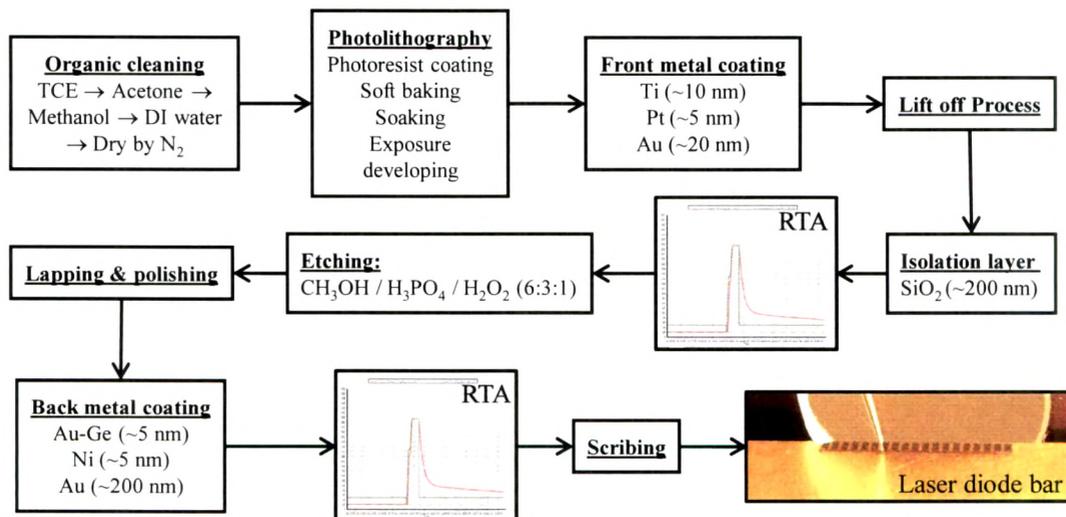


Figure 3.13: A complete procedure of device processing.

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